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Research paper

# Ni/CeO<sub>2</sub> based catalysts as oxygen vectors for the chemical looping dry reforming of methane for syngas production



Axel Löfberg\*, Jesús Guerrero-Caballero, Tanushree Kane, Annick Rubbens, Louise Jalowiecki-Duhamel\*

Univ. Lille, CNRS, Centrale Lille, ENSCL, Univ. Artois, UMR 8181, UCCS, Unité de Catalyse et Chimie du Solide, F-59000 Lille, France

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#### ABSTRACT

Chemical looping dry reforming of methane (CLDRM) is performed by exposing a Ni/CeO $_2$  solid to CH $_4$  and CO $_2$  in a cyclic way. The solid acts as an oxygen vector producing syngas (CO+H $_2$ ) during exposure to CH $_4$ , and is re-oxidized during exposure to CO $_2$ . Absence of CO $_2$  during syngas production allows suppressing reverse water gas shift reaction and reaching high selectivity. Exposure to CO $_2$  restores the oxygen capacity of the support and removes residual carbon formed at the surface, thus fully regenerating the catalyst. Solids were characterized by TPR, XRD, Raman scattering, and XPS. Results show that part of the Ni is reduced and remain in metallic state during the looping process. On the other hand, Ni $^{2+}$  species in strong interaction with Ce cations are observed even after exposure to methane. Both Ni species play important roles on reactants activation and oxygen supply by the solid. Ni loading is a crucial parameter for controlling the reduction behavior of the support and therefore for CLDRM process optimization.

 $CH_4 \rightarrow C + 2H_2$   $\Delta H^{\circ} = 75 \text{ kJ/mol}$ 

 $2CO \rightarrow C + CO_2$   $\Delta H^{\circ} = -171 \text{ kJ/mol}$ 

deactivation.

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(II)

(III)

#### 1. Introduction

The valorization of methane into liquid fuels or high hydrocarbons has attracted renewed attention in the last years [1] due to two main factors, the large volume of methane resources available as compared to petroleum, particularly since the exploitation of shale gas has been developed [2,3], and the remarkable contribution to the greenhouse effect of this molecule.

Combining the valorization of  $CH_4$  with that of  $CO_2$  through the dry reforming of methane (DRM, Eq. (I)) brings the advantage of consuming another gas which contributes strongly to the global warming. DRM produces  $H_2$  and CO, which jointly or separately, are the raw materials for different processes producing energy and chemicals, e.g. through Fischer-Tropsch synthesis.

$$CH_4 + CO_2 \rightarrow \ 2H_2 + 2CO \quad \Delta H^\circ \ = \ 247 \ kJ/mol \tag{I} \label{eq:local_equation}$$

Among the difficulties is the formation of coke due to the decomposition of methane into carbon and hydrogen (Eq. (II)) [4] or by the disproportionation of carbon monoxide (Boudouard reaction, Eq. (III)) [5] at the high operating temperatures. In both cases, this leads to the formation of carbon and to the probable deactivation of the catalyst. Sintering of the active species at the high tempera-

Another drawback of DRM is due to the simultaneous presence of  $CO_2$  (reactant) and hydrogen (product) leading, by the reverse water gas shift (RWGS, Eq. (IV)) reaction, to a loss in selectivity.

tures needed to perform such reaction is another cause of catalysts

$$CO_2 + H_2 \rightarrow CO + H_2O \quad \Delta H^\circ = -41 \text{ kJ/mol} \tag{IV} \label{eq:energy}$$

One solution to overcome these drawbacks consists in separating the reactant and performing DRM in a so-called chemical looping process. It consists in reacting methane with an oxygen vector material that would get reduced and produce selectively syngas (Eq. (V)). In a second time, the reduced solid would be re-oxidized by  $\rm CO_2$  to regenerate the oxygen vector (Eq. (VI)). The process globally corresponds to the catalytic DRM although in practice it consists of two independent gas-solid reactions (Fig. 1).

$$CH_4 + Sol - O \rightarrow 2H_2 + CO + Sol - R \tag{V}$$

$$CO_2 + Sol - R \rightarrow Sol - O + CO \tag{VI}$$

Sol-O = oxidizedcatalystandSol-R = reducedcatalyst

Although the overall process would remain endothermic as in co-feed DRM, chemical looping would bring several advantages mainly in terms of catalysts deactivation and selectivity as, first,

E-mail addresses: Axel.Lofberg@univ-lille1.fr (A. Löfberg), Louise.Duhamel@univ-lille1.fr (L. Jalowiecki-Duhamel).

<sup>\*</sup> Corresponding authors.

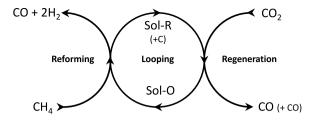


Fig. 1. Illustration of CLDRM process.

the deactivation by carbon is avoided as the solid is regenerated in each cycle and deposited carbon can be re-oxidized by  $CO_2$  (reverse Eq. (III)). Second, the hydrogen produced in the first step is never in contact with the carbon dioxide, thus avoiding the RGWS (Eq. (IV)).

Several industrial techniques have been already applied since decades in order to improve reaction systems on this way [6–9]. The main difference from co-feed processes is the ability of decoupling the reaction in two. Therefore, the operation conditions can be optimized independently, obtaining in some cases improved performances in respect to that reached in co-feed.

To some extent this approach is similar to that proposed for the redox decoupling of Mars and Van Krevelen reactions, e.g. for butane selective oxidation to maleic anhydride [10,11]. It should be noted however that such mechanism has not been proved for DRM although Sadykov et al. have suggested it may apply in the case of fluorite and perovskite supported metal nanoparticles [12].

One of the first researchers to propose such system for DRM for syngas production was Otsuka et al. [13] in 1998 on Pt-CeO<sub>2</sub> based catalyst.

Bhavsar et al. [14–17] proposed a concept for the combustion of methane and the production of CO consisting of a variant of chemical looping combustion (CLC) using CO<sub>2</sub> instead of O<sub>2</sub>. Although the production of hydrogen was not the goal, they defined the process as a "chemical looping dry reforming of methane". They essentially focused on Fe-CeO<sub>2</sub> systems which indeed do not lead to selective oxidation in syngas but to total combustion to CO<sub>2</sub> and water. CO is essentially produced during the regeneration step. They discussed about the relevance of the oxidation state of iron towards the selectivity of the products. The distribution goes from a total oxidation of CH<sub>4</sub> (from Fe<sub>2</sub>O<sub>3</sub> to Fe<sub>3</sub>O<sub>4</sub>), going through a mix of total-partial oxidation to syngas (from Fe<sub>3</sub>O<sub>4</sub> to FeO) and ending with the formation of carbon when the reduction of the solid is complete (Fe°). Even though the oxidation by CO<sub>2</sub> gives another "initial" state of Fe species (Fe<sub>3</sub>O<sub>4</sub>) for the next cycle, the system is still not 100% selective towards H<sub>2</sub> and CO. This will certainly depend on the nature of the oxygen vector but can also depend on the diffusion rate of oxygen which needs to be optimal for the partial oxidation. If the rate is too high the total oxidation of methane can take place producing H<sub>2</sub>O and CO<sub>2</sub>, on the contrary, if it is too slow carbon deposition can be produced from the decomposition of methane. Zhu et al. have also proposed the use of CeO2-Fe2O3 materials as oxygen carriers for the steam reforming of ethane in chemical looping systems [18].

More recently, Galvita et al. [19] went further in such concept by proposing Ni promoted Fe-CeO<sub>2</sub> solids for the same application with relevant increases in performances but still aiming towards total oxidation and not syngas production. Similarly, Bhavsar studied NiFe-CeO<sub>2</sub> systems in a looping process mostly focusing on the oxidation step for CO<sub>2</sub> valorization to CO [17].

In the case of methane reforming for syngas production, the main challenges are (i) to obtain the partial oxidation of methane without neither CO<sub>2</sub> nor H<sub>2</sub>O during the first step, (ii) possibly using transition metals-based catalysts and (iii) keeping a high activity towards methane while producing as little carbon deposition as possible. This last point is less crucial than the previous considering that eventual carbon deposits may be removed during the

regeneration step. It can be worthy nevertheless to limit such carbon deposit as much as possible. Indeed, in another proposal for periodic reforming of methane with  $CO_2$ , S. Assabumrungrat et al. [20] suggested to use Ni/SiO<sub>2</sub>-MgO based catalysts to perform the decomposition of methane in carbon and hydrogen in a first step and the combustion of this carbon by  $CO_2$ , producing  $CO_3$ , in a second. In this case, the solid does not act as an oxygen vector. Although the results were promising they showed that a certain amount of oxygen needed to be added in the  $CO_2$  stream in order to achieve full removal of carbon.

Very recently, Tang et al. proposed a comprehensive review of the state of the art of chemical-looping process for reforming of methane [21], with a particular attention toward the oxygen vector materials. One of the key points is the possible re-oxidation by  $CO_2$  instead of  $O_2$ .

 $CeO_2$  is well known for its oxygen mobility and storage capacity. From the thermodynamic point of view, the oxidation of reduced ceria by CO<sub>2</sub> is favorable at 25 °C up to more than 1000 °C [20]. Otherwise, the main constrains come from the slow kinetics at low temperatures, even using noble metals to accelerate the process. Li et al. [22] attributed the activation of CO<sub>2</sub> on CeO<sub>2</sub> to the rapid formation of carbonates and carboxylates species on partially reduced  $CeO_{2-x}$ , which could remain on the surface of the solid up to 500 °C. However, some other studies concluded that CO<sub>2</sub> can be directly dissociated to CO and active oxygen species by the metal-ceria catalyst, donating oxygen to a lattice vacancy and producing CO, which is totally desorbed at around 450 °C [23,24]. In addition, Jin et al. [25] observed in a Temperature Programmed Desorption of CO<sub>2</sub> on Pt/CeO<sub>2</sub>, the production of CO around 280 °C (the catalyst was pre-reduced under H<sub>2</sub>). Moreover, Sharma et al. [26] noted that the oxidation of ceria by CO<sub>2</sub> was negligible in measurements using Pd/ceria at 200 °C, implying that indeed there are kinetic limitations to its oxidation by carbon dioxide. Nevertheless, they demonstrated that at 350 °C the catalyst is somehow oxidized. CO was produced when a pulse of CO<sub>2</sub> was fed on the reduced CeO<sub>2</sub> (the CeO<sub>2</sub> was pre-reduced by CO). In addition, O. Demoulin et al. [23] showed that pure  $CO_2$  is capable to oxidize reduced (under  $H_2$ )  $Pd/Ce_{0.21}Zr_{0.79}O_2$ catalyst at 300 °C. Nevertheless, the key factor is again the kinetics of the oxidation as 34 min are necessary to oxidize completely the solid (instead of 12 min under  $O_2$ ). Otherwise, Staudt et al. [27] claimed that CO<sub>2</sub> is activated in the defect sites of ceria, even without noble metals, if the concentration of Ce<sup>3+</sup> is high enough. Otsuka et al. [13] showed almost total oxidation of CeO<sub>2</sub> by CO<sub>2</sub> at 450 °C with the formation of CO (after reduction by CH<sub>4</sub> at 700 °C), getting a recovery degree of about 82%.

As in co-feed DRM, efforts should be focused on developing processes with transition metals instead of expensive Pt. Although Ni/CeO<sub>2</sub> catalysts have been thoroughly studied for reforming reactions, they have, to the best of our knowledge, never been considered for chemical looping dry reforming of methane for syngas production.

Therefore, we propose to explore the properties of Ni/CeO<sub>2</sub> as oxygen vector for this reaction. Before developing the experimental aspects an overview of the thermodynamics of such system is necessary. Results obtained for such Ni/CeO<sub>2</sub> based catalysts prepared by impregnation are then presented and discussed on the basis of different experimental parameters such as reaction temperature or Ni loading.

#### 2. Experimental

#### 2.1. Catalysts preparation

All the catalysts were prepared by precipitation of CeO<sub>2</sub> and wetness impregnation of the Ni on the homemade support. Ceria

oxide catalyst support was synthesized using triethylamine (TEA)  $(C_2H_5)_3N$ , (Sigma-Aldrich,  $\geq 99.5\%$  assay) as the precipitant agent. As an example of this synthesis, 21.9 g of cerium (III) nitrate hexahydrate CeN $_3O_9$ ·6H $_2O$  (Fluka, >99% assay), was dissolved into 0.1 L distilled H $_2O$ . Separately, 0.1 L of methanol CH $_3OH$  (Sigma-Aldrich,  $\geq 99.8\%$  assay), as the media agent, was mixed with 0.0278 L of TEA. The solution of cerium nitrate hexahydrate was added slowly (around 1drop/s, vigorous stirring) into the mixing of methanol and TEA. Once the addition was completed, the stirring was kept during 2 h more. The obtained solid was washed with distilled water and methanol. The product was filtered under vacuum and then dried during 24 h at 100 °C. The solid was calcined at 500 °C during 4 h in air with a heating ramp of 5 °C/min.

Nickel was loaded to ceria by the wet impregnation method. In this case, Ni(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O (Sigma-Aldrich,  $\geq$ 98.5%) was used as a precursor. The proper amount of Ni(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O was dissolved in water, then 1 g of ceria was added to it. The mixture was stirred during 1 h. Afterwards, the water was evaporated at 60–70°. Then, the solid was removed from the beaker and dried in the oven during 1 h at 100 °C. Finally, the catalyst was treated by calcination at 500 °C during 4 h in air with the heating ramp of 5 °C/min. Three ranges of Ni loading were considered: low (2.1 wt%), intermediate (8–11 wt%) and high (38.5 wt%). In the intermediate range results shown concern three different batches with respectively 8.8, 9.5 and 10.5 wt% of Ni.

#### 2.2. Catalysts characterization

The crystalline phases of the catalysts were measured by X-ray diffraction (XRD) using a Bruker D8 Advance X-ray diffractometer equipped with a fast detector type LynxEye with a copper anticathode. The analysis was made in the  $2\theta$  domain (20– $90^{\circ}$ ) with a step of  $0.02^{\circ}$  and time integration of 0.3s. The average crystallites sizes of CeO<sub>2</sub>, NiO and Ni<sup>0</sup> were estimated from the XRD patterns by Scherrer equation.

 $N_2$  physisorption at 77 K data were collected on a multipoint and monopoint equipment to obtain the surface area of the different catalysts before and after test.

The metal loadings for all the catalysts were analyzed by X-ray fluorescence (XRF) or inductively coupled plasma (ICP) techniques.

The behavior under  $H_2$  of these catalysts was also studied by Temperature Programmed Reduction (TPR) on a Micromeritics Autochem II Chemisorption analyzer, the  $H_2$  consumed was analyzed by a TCD detector. 25 mg of sample was treated under a flow rate of  $50\,\text{mL/min}$  (5%  $H_2/N_2$ ). The temperature was increased at a rate of  $10\,^{\circ}\text{C/min}$  up to  $900\,^{\circ}\text{C}$ .

The X-ray photoelectron spectroscopy (XPS) analyses were performed using a Kratos Analytical AXIS Ultra DLD spectrometer. A monochromatized aluminum source (AlK $\alpha$  = 1486.6 eV) was used for excitation. The X-ray beam diameter is around 1 mm. The analyzer was operated in a constant pass energy of 40 eV using an analysis area of approximately  $700\times300~\mu m$ . Charge compensation was applied to compensate for the charging effect occurring during the analysis. The C 1s (285.0 eV) binding energy (BE) was used as internal reference. The spectrometer BE scale was initially calibrated against the Ag  $3d_{5/2}$  (368.2 eV) level. Pressure was in the  $10^{-10}$  Torr range during the experiments. Simulation of the experimental photopeaks was carried out using Casa XPS software. Quantification took into account a nonlinear Shirley background subtraction.

Raman spectra were recorded at room temperature with the 647.1 nm excitation line from a Spectra Physics krypton ion laser with 3 mW laser power at the sample. The beam was focused on the compounds using the macroscopic configuration. The scattered light was analyzed with an XY 800 Raman Dilor spectrometer equipped with an optical multichannel detector (liquid

nitrogen-cooled charge coupled device). The spectral resolution was approximately  $0.5\,\rm cm^{-1}$  in the investigated  $200-1000\,\rm cm^{-1}$  range.

## 2.3. Catalytic chemical looping dry reforming of methane (CLDRM)

The catalytic test was conducted under atmospheric pressure in a fixed-bed quartz reactor fitted in a programmable oven.  $50-200\,\mathrm{mg}$  of sieved solid were mixed with SiC to keep the catalytic bed volume constant. The gas stream was switched periodically between neutral gas (Ar 100%) to one containing CH<sub>4</sub> ( $5\,\mathrm{vol}\%$ ) named "reductant step" or CO<sub>2</sub> ( $5\,\mathrm{vol}\%$ ) named "oxidant step". He ( $5\,\mathrm{vol}\%$ ) was introduced together with CH<sub>4</sub> and CO<sub>2</sub> in order to allow internal calibration of the reactant flows. The total flow rate was maintained constant ( $F_T = 100\,\mathrm{mL/min}$ ). This kind of method is one of the easiest ways of doing such process in the laboratory scale. The outlet gases were analyzed online by a mass spectrometer (QMS200 from Pfeiffer).

Typically, each cycle consisted in exposing the sample to CH<sub>4</sub> for 1 min, Ar for 2 min, CO<sub>2</sub> for 1 min and again Ar for 2 min. Cycles were repeated 12 or 60 times. The conversion of each reactant was integrated on a full cycle as well as the production of products. Ideally the conversions of CH<sub>4</sub> and CO<sub>2</sub> should be equal. The H<sub>2</sub>/CO ratio reported is the one measured during the reductant step and should ideally be equal to 2. Other relevant values are reported for each cycle. In particular, the  $C_{deposited}/C_{converted}$  ratio corresponding to the proportion of converted methane transformed in carbon during the reductant step. This value is obtained by calculating the mass balance of all carbon containing species.  $CO/(CO_2 + CO)$  ratio in the reductant step is indicative of the selectivity towards partial oxidation. Ideally, it should be equal to 1 and lower values are indicative of total oxidation. Finally, CO/CO<sub>2converted</sub> ratio in the oxidant step is indicative of eventual carbon deposit oxidation (>1), ideally it should be equal to 1. All data are obtained with an error margin of 5%<sub>rel</sub>.

#### 3. Results and discussion

#### 3.1. Thermodynamics

The first step of such study consists in determining the potential constituents of solids based on thermodynamics regarding the feasibility of the oxidation and reduction by  $CO_2$  and  $CH_4$  respectively.

Dry reforming of methane is an endothermic reaction which needs to be carried out at high temperature, typically above 600 °C and it can be limited by the thermodynamic equilibrium. Furthermore, as mentioned, other reactions may take place in these conditions, such as RWGS, methane cracking or the Boudouard reaction. Thermodynamics of classical co-feed DRM are well known and have been extensively reported in the literature [e.g. 2]. We will thus limit the discussion to those elements which are relevant to the understanding of the periodic behavior of the system.

The set of the standard Gibbs free energy was calculated using the "factsage.com" [28] website database.  $\Delta_f G^{\circ}$  at 600 °C, 700 °C and 800 °C for several relevant reactions are reported in Table S1.

DRM (reaction (1) in Table S1) becomes favorable around 650 °C. Cracking of methane is more favorable than DRM up to 725 °C, which is one of the reasons about obtaining less carbon at high temperatures. In addition, the oxidation of the deposited carbon with CO<sub>2</sub> (2) is favorable above 700 °C but more difficult to occur thermodynamically than carbon deposition (3) in the whole working range. Therefore, the eventual carbon would be quite difficult to oxidize at the same time as the DRM occurs due to the competition

between  $CH_4$  and C to react with  $CO_2$ , as the one of  $CH_4$  is more favorable.

Moreover, the absolute  $\Delta_f G^\circ$  of RGWS **(4)** is very small in all the temperature range. Thus, 100% selectivity towards syngas is quite difficult to reach from the thermodynamic point of view in the classical DRM process.

The chemical looping dry reforming of methane (CLDRM) system needs particular properties. The solid plays a role as a reactant of the reaction (besides the catalytic properties of activating CH<sub>4</sub> and CO<sub>2</sub>, in this case separately). The following calculations are based on known solid phases for which thermodynamic data is available. For ceria, two redox systems (CeO<sub>2</sub>–Ce<sub>6</sub>O<sub>11</sub> and CeO<sub>2</sub>–Ce<sub>2</sub>O<sub>3</sub>) are considered.

#### 3.1.1. Reactivity of Ni species

Table S1 shows that the total oxidation of methane towards H<sub>2</sub>O and CO<sub>2</sub> (7) is the most favorable to occur within the different possible pathways to reduce NiO (5, 6 and 7). Unless the selectivity is somehow controlled by kinetic means, this shows that, at first sight, Ni-NiO system is not adapted to act as an oxygen vector for CLDRM.

On the other side, Table S1 shows that the oxidation of Ni<sup>0</sup> to NiO by CO<sub>2</sub> (8) is not favorable in the temperature range explored. This is a crucial property from the selectivity point of view. Indeed, the unselective reduction of NiO towards Ni producing H<sub>2</sub>O and CO<sub>2</sub> would occur only once, at the beginning of the experiment. On this way, Y.H. Taufiq-Yap et al. [29] observed Ni<sup>0</sup> without any peak of NiO in the XRD patterns after DRM experiment. Obviously, the re-oxidation of Ni° by O<sub>2</sub> (9) is feasible in the whole range of temperature considered. This type of behavior can be also correlated with the literature. Bhavsar et al. [16] studied the chemical looping combustion (thus re-oxidizing the catalyst with air) over Ni-CeO<sub>2</sub>. They showed clearly the production of CO<sub>2</sub> which is in concordance with the reduction of NiO to produce water and carbon dioxide. Regarding Ni, it can therefore be deduced that except during a first reduction step to Ni°, nickel will not act as an oxygen vector for CLDRM. It may nevertheless play an important role on the reactivity point of view.

#### 3.1.2. Reactivity of Ce species

The partial oxidation of  $CH_4$  on  $CeO_2$  to  $Ce_2O_3$  (10) is more favorable than the total oxidation (11 and 12). Moreover, the selective oxidation of methane between  $CeO_2$  and  $Ce_6O_{11}$  (13) is even more favorable than the direct reduction of  $CeO_2$  to  $Ce_2O_3$  (10). In this context, Otsuka et al. [13] showed that the oxidation of  $CH_4$  over  $CeO_2$  becomes feasible at temperatures above 650 °C, which is quite close to our calculations (680 °C for reaction (13)). In all cases the selective oxidation of methane towards syngas is more favorable than total oxidation, making  $CeO_2$  a good candidate to act as oxygen vector in CLDRM.

Obviously, the other main reaction is the re-oxidation of  $Ce_xO_y$  by  $CO_2$  (14, 15, 16), which in all the cases is thermodynamically favorable at the working temperature range (600–800 °C). Moreover, the  $Ce_xO_y$  oxidation is more feasible at lower temperature. Nevertheless, the variation of the Gibbs free energy on the oxidation of  $Ce_6O_{11}$  to  $CeO_2$  is really close to zero, meaning that some thermodynamics limitations can be expected in this case. As a reference, Otsuka et al. [13] demonstrated that the reduced ceria on a Pt- $CeO_2$  catalyst can be oxidized by  $CO_2$ . Otherwise, NiO and  $Ce_xO_y$  (17, 18) are always in favor of Ni species reduction, which should maintain Ni in metallic state if no  $O_2$  is present. Furthermore, it can be seen that ceria will not oxidize carbon eventually deposited at its surface (9, 20).

In summary, Ni-CeO<sub>2</sub> system appears to be a very good candidate for Chemical Looping DRM although it is known to be a good catalytic solid towards CLC. The main difference is that in the case

of the re-oxidation by  $CO_2$ , NiO cannot be formed and  $CeO_2$  acts as an unique potential oxygen vector which is selective to syngas production.

It must be underlined that such thermodynamic considerations are based on known and stable phases for which thermodynamic data is available. One cannot exclude the existence, in the working conditions, of other phases with reactivities in favor or disfavor of CLDRM.

#### 3.2. Catalysts characterization

The Table 1 summarizes the main properties of the  $Ni/CeO_2$  catalysts.

The X-ray diffraction patterns of the different Ni loaded compounds (Fig. 2) evidence characteristic fluorite structure of CeO<sub>2</sub> in all the cases. NiO is not observed at the lowest nickel content which could be explained by several reasons: (i) the particles may be too well dispersed on CeO<sub>2</sub> to be detected, (ii) a solid solution of cerium and nickel could be formed, or (iii) the particles may be amorphous. At higher loadings, narrow peaks of NiO are observed at 36.8°, 43.2°, 63.1° and 75.4° with average crystallites sizes of 21, 16 and 18 nm for Ni(38.5)/CeO<sub>2</sub>, Ni(8.8)/CeO<sub>2</sub> and Ni(9.5)/CeO<sub>2</sub>, respectively. The same average crystallites size for CeO<sub>2</sub> (12 nm) is obtained at high (38.5%) and low (2.1%) loadings of nickel. In addition the surface areas of the catalysts are similar, 27 m<sup>2</sup>/g for  $Ni(2.1)/CeO_2$  and  $33 \text{ m}^2/\text{g}$  for  $Ni(38.5)/CeO_2$  to the ones observed for  $Ni(8.8)/CeO_2$  (29 m<sup>2</sup>/g) and  $Ni(9.5)/CeO_2$  (32 m<sup>2</sup>/g). Thereby, the loading of nickel, at least even up to 38.5%, does not seem to affect the total surface area of the catalyst using the impregnation method. Furthermore, the reproducibility of this method is checked comparing the properties of the Ni(8.8)/CeO<sub>2</sub> and Ni(9.5)/CeO<sub>2</sub>, which shows almost identical XRD patterns, average crystallites sizes and surface areas. However, compared to ceria alone adding Ni leads to an increase of the average crystallites size and a decrease of the specific surface area.

The TPR of Ni(38.5)/CeO<sub>2</sub>, Ni(9.5)/CeO<sub>2</sub> and Ni(2.1)/CeO<sub>2</sub> (Fig. 3) show two peaks between 265 and 365 °C which are related to the reduction of nickel species. While the peak related to reduction of ceria is at 794 and 800 °C for 38.5 wt% Ni and 2.1 wt% Ni, respectively. It is significantly lower for 9.5 wt% Ni (766 °C). Large NiO crystallites seem to be more difficult to reduce as the main peak related to the reduction of nickel shifts from 315 °C for Ni(2.1)/CeO<sub>2</sub>, assuming nickel is present as NiO at such low content, going to 327 °C for Ni(9.5)/CeO<sub>2</sub> and 362 °C for Ni(38.5)/CeO<sub>2</sub>. The lowest is the Ni loading the lowest is the temperature of reduction  $(362\,^{\circ}\text{C} \rightarrow 38.5\,\text{wt}\%\,\text{Ni},\ 327\,^{\circ}\text{C} \rightarrow 9.5\,\text{wt}\%\,\text{Ni}\ \text{and}\ 315\,^{\circ}\text{C} \rightarrow 2.1\,\text{wt}\%$ Ni). Otherwise, the peak found around 270 °C could be attributed to the presence of Ni species in very small NiO crystallites and/or in a Ce-Ni-O solid solution [30,31]. Therefore, it appears that in all the studied compounds, there are some Ni species in strong interaction with Ce species that can be found at the interface between NiO and CeO<sub>2</sub> or in Ce-Ni-O solid solution.

#### 3.3. Catalytic performances in CLDRM

#### 3.3.1. Activity of CeO<sub>2</sub>

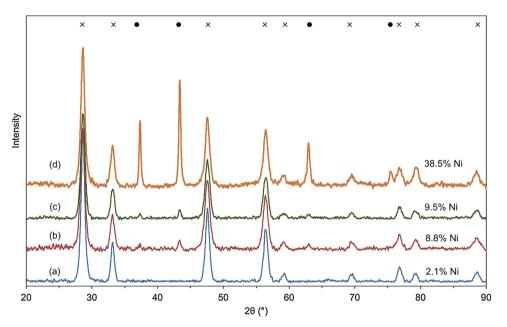
The role of the  $CeO_2$  without an active metal was investigated. The experiments were carried at 700 °C, 750 °C and 800 °C. Fig. S1 shows the partial pressures in the outlet of the reactor at 800 °C. The other two experiments are not shown as the activities were negligible.

Comparing the reaction cycles to the first one performed with the closed reactor (by-pass), it can be seen that the partial pressures of  $CH_4$  and  $CO_2$  rapidly reach the nominal value in each reductant and oxidant step. Indeed, the conversions of  $CH_4$  and  $CO_2$  are around 0.5–1%.

**Table 1**Ni loading, surface area (BET) and average crystallites size (from XRD) of the different catalysts

Catalyst	Ni loading (wt.%)	Ni/M <sub>T</sub> Atomic ratio	S <sub>BET</sub> (m <sup>2</sup> /g)	d <sub>NiO</sub> (nm)	d <sub>CeO2</sub> (nm)
CeO <sub>2</sub>	-	_	53	_	6
Ni(2.1)/CeO <sub>2</sub>	2.1	0.06	27	n.o.	12
Ni (8.8)/CeO <sub>2</sub>	8.8	0.22	29	16	11
Ni(9.5)/CeO <sub>2</sub>	9.5	0.24	32	18	10
Ni (10.5)/CeO <sub>2</sub> Ni(38.5)/CeO <sub>2</sub>	10.5 38.5	0.26 0.65	35 33	n.a. 21	n.a. 12

n.o.: not observed. n.a.: not analysed.



**Fig. 2.** XRD patterns for (a) Ni(2.1)/CeO<sub>2</sub>, (b) Ni(8.8)/CeO<sub>2</sub>, (c) Ni(9.5)/CeO<sub>2</sub>, (d) Ni(38.5)/CeO<sub>2</sub> x CeO<sub>2</sub> ● NiO.

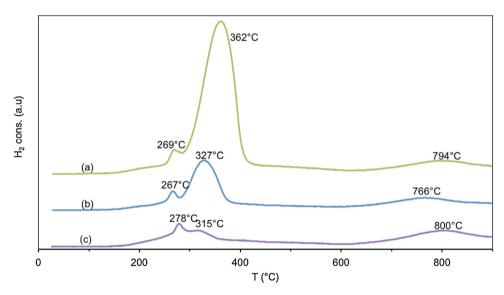


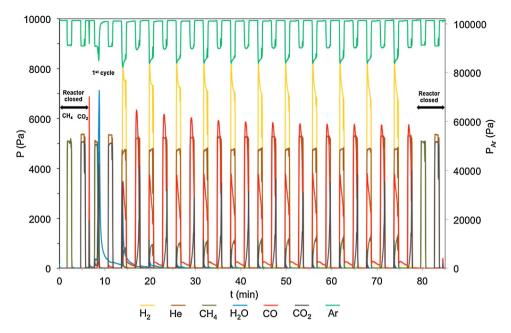
Fig. 3. TPR profile of (a) Ni(38.5)/CeO<sub>2</sub>, (b) Ni(9.5)/CeO<sub>2</sub>, (c) Ni(2.1)/CeO<sub>2</sub> under 10% H<sub>2</sub>/N<sub>2</sub> gas.

#### 3.3.2. Activity of Ni/CeO<sub>2</sub>

The partial pressures of the different compounds along the whole experiment are represented in Fig. 4 for Ni(8.8)/CeO $_2$  at 700 °C. The reactive cycles are preceded and followed by cycles with the reactor closed in order to calibrate the MS signals.

The experiment starts showing a different behavior for the first redox cycle in respect to the following ones. A considerable amount

of  $H_2O$  and  $CO_2$  is produced during the first reductant step (CH<sub>4</sub>) but decreases down to almost zero after the first 2 cycles, leading to a highly selective system towards syngas products. It should be reminded that the catalysts used were not pre-reduced before test. Therefore, the production of  $H_2O$  and  $CO_2$  in this first cycle can be ascribed to the reduction of NiO species that are subsequently not re-oxidized by  $CO_2$  as could be expected by the thermodynamics



 $\textbf{Fig. 4.} \ \ Partial\ pressures in the outlet flow working with 200 mg of Ni(8.8)/CeO_2 (0.1-0.2 mm) + 100 mg SiC at 700 °C F_{T,0} = 100 mL/min. Cycle: 1 min 5\% CH_4, and 1 min 5\% CO_2.$ 

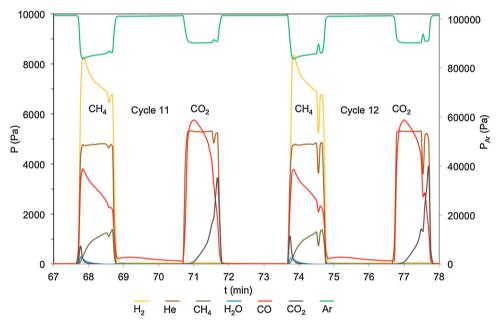


Fig. 5. Partial pressures in the outlet flow, zoom of Fig. 4.

calculations. V. Galvita et al. [32] also observed H<sub>2</sub>O and CO<sub>2</sub> in the early stage of the reaction working with Pt-Ce-Zr based catalyst on the periodic SRM. The absence of total oxidation products in the successive cycles is consistent with the hypothesis that in such reaction condition Ni° species are not oxidized to NiO. These results also confirm that CeO<sub>2</sub> may play the role of selective oxygen vector for syngas production by CLDRM as the system tends to become almost 100% selective towards H<sub>2</sub> and CO during the reductant steps with an almost negligible production of H<sub>2</sub>O and CO<sub>2</sub>. Fig. 5 shows a zoom in the last two cycles of the experiment in order to verify the excellent reproducibility from one cycle to the other.

First,  $H_2$  and CO are produced majorly during the reductant step with a really small contribution of  $H_2O$  and  $CO_2$ . The same production of  $H_2$  and CO is obtained for the two cycles show-

ing a remarkable reproducibility between the two, even after 10 cycles. In addition, the production of CO decreases progressively during the reductant step which can be due to a lack of oxygen in the carrier. However, in this experiment the amount of oxygen in CeO<sub>2</sub> (2064  $\mu$ mols) is largely enough to react with the CH<sub>4</sub> feed (205  $\mu$ mols). Even considering the redox couple between CeO<sub>2</sub> and Ce<sub>2</sub>O<sub>3</sub> where only 25% of the oxygen in CeO<sub>2</sub> is released, the oxygen available is still high (516  $\mu$ mols O) which is not the case if the reduction to Ce<sub>6</sub>O<sub>11</sub> is considered (172  $\mu$ mols O).

On the other side, the  $CO_2$  partial pressure does not reach the nominal value of 5000 Pa during the oxidant step. Probably,  $CO_2$  needs more time to oxidize completely the  $Ce_xO_y$  species at the given working conditions. As expected, there is an important production of CO during this step due to the oxidation of  $Ce_xO_y$  species

and, eventually, of some carbon deposited during the reductant step.

A graph with the main values of conversion and selectivity is shown in Fig. 6. In practice, the properties from the reductant step in the cycle 1 are shown above the number 1 along the x-axis, whereas the properties for the oxidant step in the cycle 1 are shown above the number 1.5 along the x-axis to differentiate between the two and to not get confused with classic results of DRM in co-feed.

There is an evolution of the conversion between the first cycle and the following ones. During the first cycle 28% conversion of CH<sub>4</sub> and only 9% of CO<sub>2</sub> are obtained, so relatively low conversions are obtained on the oxidized compound. Then, the two become stable around 80% until the end of the experiment, showing an activation phenomenon with increasing conversions. Besides, H<sub>2</sub>/CO ratio goes from 3.2 in the first cycle to 2.2 in the last one. Even though  $H_2/CO$  is equal 3.2, the ratio  $CO/(CO_2 + CO)$  is almost zero, meaning that there is no or negligible production of CO. Therefore, the H<sub>2</sub>/CO value at that first cycle is not relevant as it would come from some residual production of CO. Otherwise, the selectivity between H<sub>2</sub>, CO, H<sub>2</sub>O and carbon deposit become more coherent from the cycle 2. Indeed, the CO/(CO<sub>2</sub>+CO) ratio goes close to 1 (0.98), which corroborates the small production of CO<sub>2</sub>. The C<sub>deposited</sub>/C<sub>conv.</sub> (%) decreases from 15% in the cycle 2 to 8% in cycle 12, at the same time the  $H_2/CO$  ratio decreases from 2.5 to 2.2. This trend is coherent considering that if less cracking of methane occurs in the system, less production of hydrogen would come from this reaction. In addition, the low deposition of carbon during the reductant step is also corroborated thanks to the CO/CO<sub>2,conv</sub> ratio obtained in the oxidant step, which goes around 1.1. Such value means that almost all the CO<sub>2</sub> oxidizes the ceria species and only a small amount is used to oxidize the carbon deposit.

Moreover, Fig. S2 shows the instantaneous conversions in order to follow the behavior of the reactants within the step.

The conversion of  $CH_4$  goes from 100% to around 70% evidencing the dynamic behavior of the periodic process. Probably, the kinetic is faster when ceria is totally oxidized and then it is slower as the oxygen reservoir starts to decrease. In addition, obtaining 100% conversion at the beginning of the reductant step in each cycle reinforces the advantage of working in a dynamic process from the thermodynamic point of view, as the limitations are more difficult to reach than working in co-feed (stationary state). From the other side, the  $CO_2$  conversion goes almost to zero at the end of the oxidant step. This trend is favorable for our interest.  $CO_2$  conversion going to zero would mean that all the species are re-oxidized and the use of  $CO_2$  is close to optimal. Therefore, this experiment is a good opportunity to compare the mass balance on the catalyst towards oxygen and carbon in the whole cycle.

The Fig. S2 also shows that the overall approx. 80% conversions of  $CO_2$  and  $CH_4$  (Fig. 6) are not due to thermodynamic limitations as would occur in classical DRM. Indeed, if the steps had been shorter, the integrated conversion would have given higher values, eventually reaching 100%. This is an important point to take into account in comparing classical DRM performances to CLDRM.

Table 2 shows the mass balance of oxygen and carbon in the catalyst during the two steps of the last four cycles. The amount of oxygen released in the reductant step is near to that inserted during the oxidant step. In addition, the amounts of carbon deposited during the reductant step correspond to those oxidized during exposure to CO<sub>2</sub> this indicates that carbon does not accumulate on the surface of the solid in these experimental conditions. Mass balances thus confirm the hypothesis about CO<sub>2</sub> being capable to oxidize both the reduced ceria and the eventual carbon deposit in such conditions. In an overall view, this experiment has shown the principles and the procedures that we follow to process and understand the data of the different experiments in periodic. Besides, the experiment shown demonstrates the feasibility of working in

such conditions even at 700 °C, obtaining promising conversions and selectivity.

Supporting Ni species on CeO<sub>2</sub> allows drastically increasing CH<sub>4</sub> and CO<sub>2</sub> conversions as conversions of about 1% were obtained on CeO<sub>2</sub> alone (Fig. S1). Clearly, Ni species have an essential role in the activation of the reactants. Besides, CeO<sub>2</sub> playing the role of selective oxygen vector for syngas production furnishes the O species which leads to the creation of anionic vacancies in the solid. Activation of CH<sub>4</sub> on Ni species is expected, however CO<sub>2</sub> activation that has been proposed on anionic vacancies [33–35] is also largely enhanced in the presence of Ni species. This could be explained by the presence of Ni species in strong interaction with Ce species increasing the number of anionic vacancies in CeO<sub>2</sub> and/or in the Ce-Ni-O solid solution.

#### 3.3.3. Stability test

A 60 cycles experiment instead of 12 were carried at  $800\,^{\circ}$ C on Ni(8.8)/CeO<sub>2</sub> in order to check the stability of the process (Fig. 7). The conversions of CH<sub>4</sub> and CO<sub>2</sub> reach 95% and 99%, respectively. In addition, around 100% selectivity is found obtaining H<sub>2</sub>/CO=2.0, CO/(CO<sub>2</sub>+CO)=0.96 and C<sub>deposited</sub>/C<sub>conv.</sub>=4%. The performance underlines clearly the feasibility of DRM in periodic conditions with the advantages associated, such as avoiding the RWGS. Interestingly the proportion of coke formed diminishes along the cycles.

## 3.3.4. Characterization of catalysts after reduction and oxidation steps

In order to characterize the state of the catalyst and eventual differences after the reduction and oxidation steps, a series of four separate experiments were performed using 200 mg of Ni(10.5)/CeO<sub>2</sub>. CLDRM was performed at two different temperatures (600 °C and 800 °C) and stopped at the convenient moment (i.e. after a reduction or oxidation step) after 30 cycles in order to be sure that the solid is "stable" in terms of the internal evolution. Samples were cooled down to room temperature in flowing argon.

3.3.4.1. XPS. Cerium compounds have XPS spectra with rather complex features due to numerous initial and 4f electronic configurations. The 3d spectrum, registered on pure ceria, can be resolved into three spin-orbit doublets,  $3d_{3/2}$ - $3d_{5/2}$ , noted (u, v), (u", v") and (u"', v"'). Fig. S3 shows the Ce  $3d_{3/2.5/2}$  spectrum, observing the typical pattern of Ce<sup>4+</sup> species before experiment. The characteristic peak of Ce<sup>4+</sup> is observed at 916.7 eV. Typically, this peak does not appear for the Ce<sup>3+</sup> spectrum [36,37]. Such peak appears in all the cases after experiment. On the other hand, typical peaks for Ce<sup>3+</sup>  $(u^{\circ}, v^{\circ}, u' \text{ and } v' \text{ at } 899.1, 880.9, 903.4 \text{ and } 885.2 \, eV, \text{ respectively})$ are absent from all the XPS spectra. Therefore, the surface CeO2 is apparently not reduced to Ce<sup>3+</sup>. This could be explained by a rapid diffusion of oxygen species from the bulk towards the surface of the solids during the cooling of the reactor. Indeed, surface Ce<sub>x</sub>O<sub>v</sub> can be slowly oxidized even feeding only noble gases, just because of the oxygen diffusion from the bulk [22]. A long reduction by methane or/and an in-situ XPS experiment seem to be necessary to reach Ce<sup>3+</sup> species on the first layers of the catalyst.

Fig. 8 gathers the XPS spectra for Ni 2p, O 1s and C 1s for all the samples, whereas the corresponding quantitative data is presented in Table 3.

Compared to the initial state, a decrease in Ni concentration (Table 3) is seen on all samples after reaction which can be attributed to particles sintering. Indeed, the surface area of the samples decreases drastically from approx.  $32\,\mathrm{m}^2/\mathrm{g}$  to approx.  $2\,\mathrm{m}^2/\mathrm{g}$  after test. Partial coverage of Ni species by carbon could also explain such lowering. C is observed on all samples including the calcined catalysts. Part of this C can thus be due to contamination in handling or in the analytical instrument, and it is usually used as an inter-

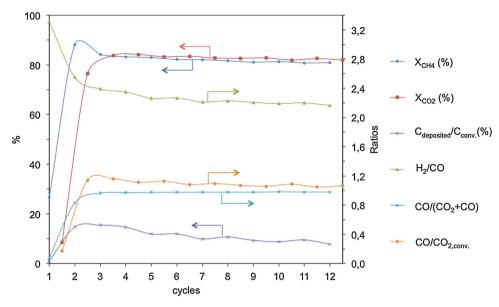


Fig. 6. Reductant step: Conversion of CH<sub>4</sub> (%),  $C_{deposited}/C_{conv.}$  (%),  $H_2/CO$  and  $CO/(CO_2 + CO)$  ratios. Oxidant step: Conversion of  $CO_2$  (%) and  $CO/CO_{2,conv.}$  ratio. 200 mg of Ni(8.8)/CeO<sub>2</sub> (0.1–0.2 mm) + 100 mg SiC at 700 °C  $F_{T,0}$  = 100 mL/min. Cycle: 1 min 5% CH<sub>4</sub>, and 1 min 5% CO<sub>2</sub>.

**Table 2**Catalyst mass balance of oxygen and carbon for the last four cycles of the experiment.

Cycle n.	Oxyg	Oxygen mass balance ( $\pm 5\%$ $\mu$ mols)			Carbon mass balance ( $\pm 5\% \mu mols$ )			
	Reductant	Oxidant	Full cycle	Reductant	Oxidant	Full cycle		
9	-152	165	13	15	-13	2		
10	-153	152	-1	14	-17	-3		
11	-151	166	15	15	-11	5		
12	-154	158	3	13	-13	0		

Mass balances: positive = uptake by solid, negative = supply by solid.

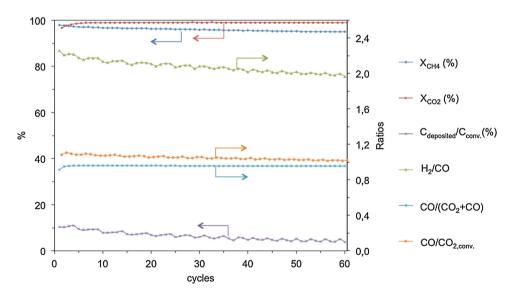


Fig. 7. Reductant step: Conversion of CH<sub>4</sub> (%),  $C_{deposited}/C_{conv.}$ (%),  $H_2/CO$  and  $CO/(CO_2 + CO)$  ratios. Oxidant step: Conversion of  $CO_2$  (%) and  $CO/(CO_{2,conv.}$  ratio. 200 mg of Ni(8.8)/CeO<sub>2</sub> (0.1–0.2 mm) + 100 mg SiC at 800 °C  $F_{T,0}$  = 100 mL/min. Cycle: 1 min 5% CH<sub>4</sub>, and 1 min 5% CO<sub>2</sub>.

nal reference. Nevertheless, surface carbon concentration increases after reaction, in particular at  $600\,^{\circ}$ C. Logically, the carbon concentration decreases after the  $CO_2$  step. One can expect carbon to be formed preferentially in proximity of Ni species affecting the XPS response of Ni. Furthermore, graphitic carbon can be identified at  $600\,^{\circ}$ C. The relative amount of such graphite type carbon is higher

after  $CO_2$  exposure indicating that such species are more difficult to remove at this temperature.

Ni  $2p_{3/2}$  XPS spectra show three main contributions. The peak at  $852.7\,\text{eV}$  observed on catalysts after test can be attributed to Ni<sup>0</sup> species. Two Ni<sup>2+</sup> species can be seen. The peak at  $853.7\,\text{eV}$  (with a multiplet split at  $855.4\,\text{eV}$ ) can be attributed to Ni<sup>2+</sup> from NiO. The contribution peak at  $855.7\,\text{eV}$  is usually attributed to Ni<sup>2+</sup>

species from Ni(OH)<sub>2</sub> hydroxides [37,38]. This is coherent with the presence of a significant contribution of O 1s at 531.5 eV observed on all samples. However, on Ni-CeO<sub>2</sub> systems, such peak at 855.4 eV has also been attributed to Ni<sup>2+</sup> species in strong interaction with Ce species [39–41].

The evolution of Ni species according to the last exposure (CH<sub>4</sub> or CO<sub>2</sub>) can be observed at both reaction temperatures. At 800 °C

the most striking evolution is the decrease after  $CH_4$  exposure, in comparison to  $CO_2$  exposure, of the  $Ni^{2+}$  from NiO contribution. Simultaneously, reduced  $Ni^0$  species increase but also the  $Ni^{2+}$  with higher binding energy at 855.7 eV which would favor the hypothesis of surface hydroxides formation and/or increase of  $Ni^{2+}$  species in strong interaction. The same trend can be observed at  $600\,^{\circ}C$ .

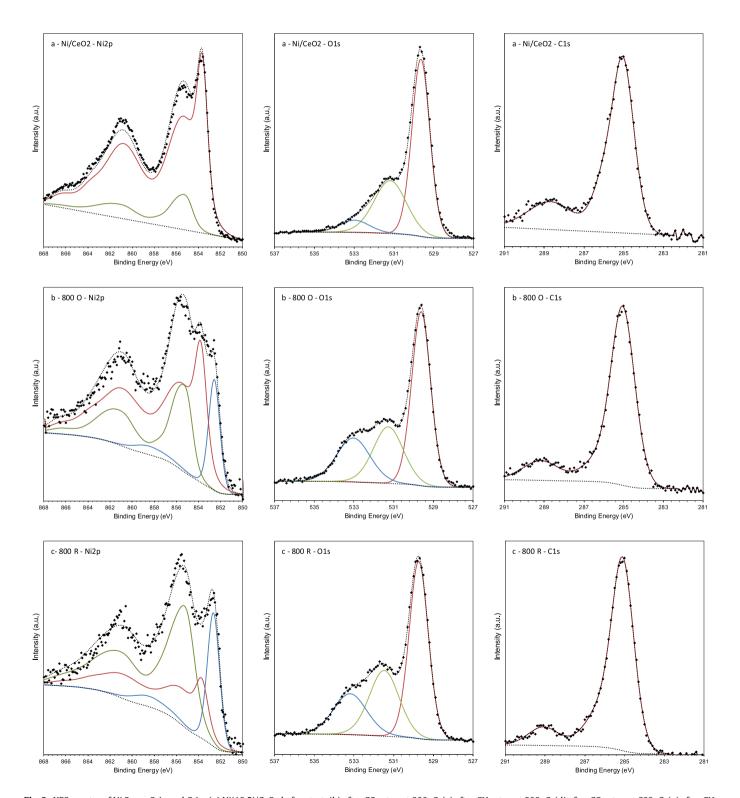


Fig. 8. XPS spectra of Ni  $2p_{3/2}$ , O 1s and C 1s: (a) Ni(10.5)/CeO<sub>2</sub> before test, (b) after CO<sub>2</sub> step at 800 °C, (c) after CH<sub>4</sub> step at 800 °C, (d) after CO<sub>2</sub> step at 600 °C, (e) after CH<sub>4</sub> step at 600 °C.

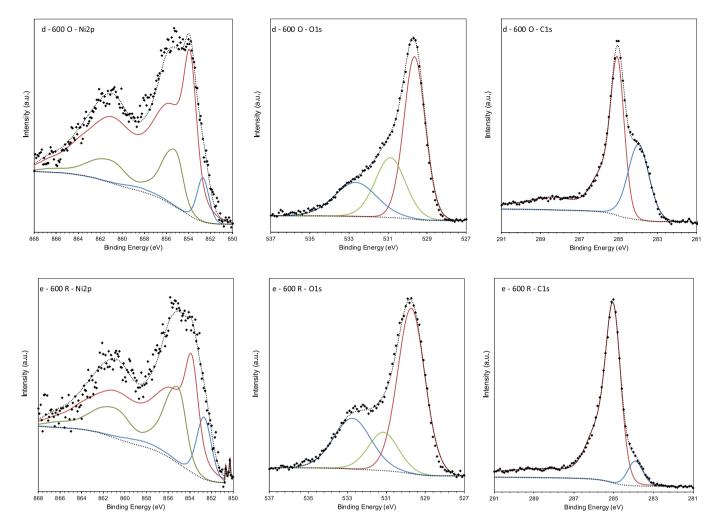


Fig. 8. (Continued)

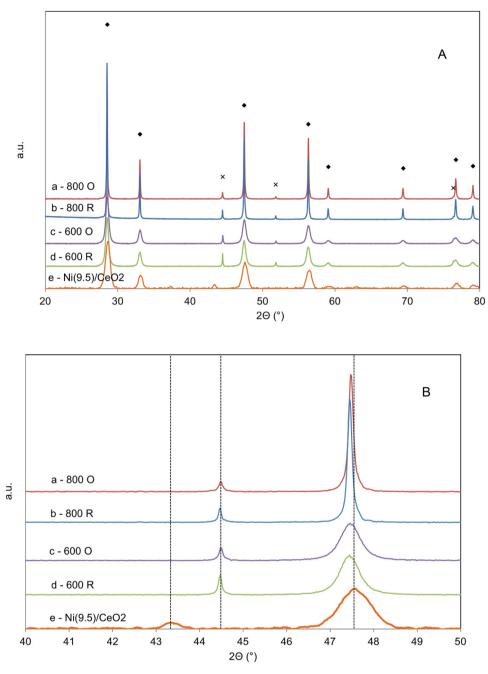
**Table 3**Data from XPS analysis for Ni(10.5)/CeO<sub>2</sub> before and after experiments.

Catalyst	Surface concentrations (at%)				Atomic ratios			
	Ce	Ni	0	С	Ni/Ce	O/Ce	C/Ce	C/Ni
Ni(10.5)/CeO <sub>2</sub>	12	24	40	23	2.1	3.4	2.0	1.0
800 °C, after CO <sub>2</sub>	11	11	42	36	1.0	3.9	3.4	3.3
800 °C, after CH₄	9	9	36	45	1.0	4.0	4.9	4.9
600 °C, after CO <sub>2</sub>	11	6	31	51	0.5	2.7	4.5	8.1
600°C, after CH₄	7	3	25	64	0.4	3.4	8.6	20.0
	Oxygen (at%)			Nickel (at%)			Carbon (at%)	
	O <sup>2-</sup> 529.7 eV	OH 531.5 eV	O' 533,2 eV	Ni <sup>0</sup> 852.7 eV	Ni <sup>2+</sup> <sub>NiO</sub> 853.7 eV	Ni <sup>2+</sup> <sub>Ni(OH)2</sub> 855.7 eV	C <sub>graph</sub> 283.9 eV	C 285.0 eV
Ni(10.5)/CeO <sub>2</sub>	55	40	6	0	64	36	-	100
800 °C, after CO <sub>2</sub>	51	26	23	13	59	28	-	100
800°C, after CH <sub>4</sub>	50	29	20	19	28	53	-	100
600 °C, after CO <sub>2</sub>	52	27	21	6	75	19	31	69
	60	15	25	11	59	30	8	92

These results also confirm [42] that surface oxidation of reduced Ni species by  $CO_2$  is possible although bulk oxidation is not thermodynamically favorable in these reaction conditions.

3.3.4.2. XRD. Figs. 9A and B shows the XRD patterns for the different samples and that of a similar starting catalyst Ni(9.5)/CeO<sub>2</sub>. For ceria, only the fluorite CeO<sub>2</sub> structure is visible. Whereas the

original catalysts show the diffraction patterns for NiO, the used samples show only Ni $^{\circ}$  confirming that in the reaction condition bulk NiO is reduced and is not re-oxidized by CO<sub>2</sub>. Fig. 9B shows in more detail the (220) diffraction peak of ceria. It can be seen that the sample reacted at 600  $^{\circ}$ C shows a slight increase of the average crystallites size (from 10 nm to 13–14 nm) and a shift towards smaller diffraction angles which are independent of the nature of



**Fig. 9.** XRD patterns (a) after CO<sub>2</sub> step at 800 °C, (b) after CH<sub>4</sub> step at 800 °C, (c) after CO<sub>2</sub> step at 600 °C, (d) after CH<sub>4</sub> step at 600 °C on Ni(10.5)/CeO<sub>2</sub> and (e) Ni(9.5)/CeO<sub>2</sub> after calcination − A:  $2\Theta = 20^{\circ}$  to  $80^{\circ}$  B:  $2\Theta = 40^{\circ}$  to  $50^{\circ}$  ( $\blacksquare$  CeO<sub>2</sub>, x Ni).

the last reaction step (CH<sub>4</sub> or CO<sub>2</sub>). After reaction at 800 °C, the samples logically show narrower diffraction peaks due to the sintering of the support (average crystallites sizes between 56 and 61 nm). Again, a slight shift to smaller diffraction angles is observed and is more significant after exposure to CH<sub>4</sub>. Such shifts towards smaller angles can be attributed to the expansion of the ceria lattice [43] as a consequence of the reduction of Ce<sup>4+</sup> (radius 0.97 Å) to Ce<sup>3+</sup> (radius 1.14 Å). No other phases of the Ce-O system can be detected.

3.3.4.3. RAMAN scattering. Fig. 10 reports the Raman spectra recorded for the 4 samples after test as well as a calcined catalyst precursor (Ni(9.5)/CeO<sub>2</sub>), the synthetized ceria support previous to Ni impregnation (CeO<sub>2</sub>) and a commercial ceria.

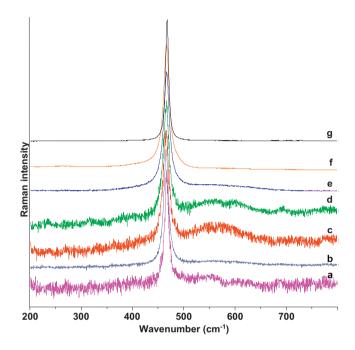
All samples show an important band at  $465\,\mathrm{cm}^{-1}$  band which is attributed to the symmetrical stretching mode ( $\nu_s(\text{Ce-O})$ ) of the

CeO<sub>8</sub> vibrational unit, environment of the cerium by the oxygen in the cubic fluorite lattice of ceria. Contrary to Fang et al. [31], no significant shift of this band can be observed between the samples which does not allow concluding on the eventual presence of a Ni-Ce-O solid solution as for very low Ni loading no shift was observed.

The two supports without Ni shows similar Raman spectra, with the exception that the 465 cm<sup>-1</sup> band is narrower for the commercial sample due to a better crystallization with respect to the prepared support. The grain size of this latter support can be estimated to 30 nm considering relation between half-width intensity and grain size established by Kosacki et al. [44]. However, the shoulder and asymmetry towards lower wavenumbers can suggest the presence of particles below 10 nm, thus in better accordance with XRD crystallites size [45–47]. A small contribution at 596 cm<sup>-1</sup> can also be detected (Table 4). This weak band is generally attributed to

**Table 4**Raman analysis for CeO<sub>2</sub>, Ni(9.5)/CeO<sub>2</sub> before experiments and Ni(10.5)/CeO<sub>2</sub> after CLDRM experiments. Particles sizes are calculated according to ref. [38]

Catalyst	$_{\nu_{S}}(\text{Ce-O})(\text{cm}^{-1})$	D <sub>e</sub> (cm <sup>-1</sup> )	D <sub>i</sub> (cm <sup>-1</sup> )	Size (nm)	Surface ratios	
					$A_{De}/A_{F2g}$	A <sub>Di</sub> /A <sub>F2g</sub>
CeO <sub>2</sub>	465.5		596.0	30		0.02
Ni(9.5)/CeO <sub>2</sub> , before DRM	465.7	547.5	615.8	24	0.20	0.12
800 °C, after CO <sub>2</sub>	464.9	547.4	615.8	76	0.14	0.08
800 °C, after CH <sub>4</sub>	466.0	547.1	615.2	59	0.62	0.17
600 °C, after CO <sub>2</sub>	465.1	542.9	600.5	27	0.48	0.37
600 °C, after CH <sub>4</sub>	464.8	542.9	600.5	24	0.56	0.36



**Fig. 10.** Raman spectra (a) after CH<sub>4</sub> step at 800 °C, (b) after CO<sub>2</sub> step at 800 °C, (c) after CO<sub>2</sub> step at 600 °C, (d) after CH<sub>4</sub> step at 600 °C on Ni(10.5)/CeO<sub>2</sub>, (e) Ni(9.5)/CeO<sub>2</sub> after calcination, (f) synthetized CeO<sub>2</sub> and (g) commercial CeO<sub>2</sub>.

 $O^{2-}$  vacancy positions [46,48] or to defects in nanosized particles. This is also referred as intrinsic defect mode [49,50].

The impregnation of Ni does not affect the position of the 465 cm<sup>-1</sup> band. The band is slightly narrower and more symmetrical suggesting a better crystallization. No NiO signal at approximately 520 cm<sup>-1</sup> could be detected because of a shadowing effect by the long tail of the first-order CeO<sub>2</sub> peak. The spectra are normalized to the height of this band. However, a significant band can be observed in the 500–630 cm<sup>-1</sup> region, which can be decomposed in two contributions at 547 cm<sup>-1</sup> and 615 cm<sup>-1</sup> (see Fig. 11). The component at 547 cm<sup>-1</sup> can be assigned to an extrinsic defect mode induced by oxygen vacancies that can be increased in case of the substitution of Ce<sup>4+</sup> by lower valence cations in a solid solution. This phenomenon has been observed for various dopants [48,49,51–53], as well as for Ni<sup>2+</sup> [54–56].

This catalyst precursor thus shows both intrinsic and extrinsic oxygen vacancies although it has been calcined. This is a good indication of strong interaction between Ni and Ce species although the typical shift of the  $\nu_s(\text{Ce-O})$  band with Ce-Ni-O solid solution formation is not observed. The ratios of peak surfaces in respect to the main  $\nu_s(\text{Ce-O})$  peak are reported in Table 4 for both intrinsic  $(A_{Di})$  and extrinsic  $(A_{De})$  defect modes.

Both samples reacted at  $600\,^{\circ}\text{C}$  show a slightly narrower  $\nu_s(\text{Ce-O})$  band at  $465\,\text{cm}^{-1}$ . Particles sizes can be estimated to 24 and 27 nm for the samples after CH<sub>4</sub> and CO<sub>2</sub>, respectively. The relative contribution of the defect bands is strongly enhanced in both

cases. In this case, both reduction and eventual substitution of  $Ce^{4+}$  by  $Ni^{2+}$  cations contribute to the generation of oxygen vacancies. The two samples show very similar spectra confirming that – at this temperature – the samples are in a more reduced state even after the  $CO_2$  reaction step.

After reaction at  $800\,^{\circ}$ C, the  $\nu_s(\text{Ce-O})$  band position is not affected by the last reacting atmosphere. The width of this band reveals a further increase of the particles size (59 and 76 nm after CH<sub>4</sub> and CO<sub>2</sub>, respectively) in accordance with XRD results. The sample exposed to CH<sub>4</sub> in the last step shows an enhanced contribution of the defect bands again showing the partial reduction of the system. Interestingly, after exposure to CO<sub>2</sub> the sample recovers a Raman spectrum very similar to the one of the calcined catalysts. This is a good indication of the regeneration of the oxygen vector in these conditions.

#### 3.3.5. Dynamic behavior of Ni/CeO2 in CLDRM

A first approach on the dynamics of the periodic process within a specific cycle is proposed. Figs. 12-14 show the cumulated amount CH<sub>4</sub> consumed and of oxygen removed from the solid along the one-minute feed of CH<sub>4</sub> at  $700\,^{\circ}$ C. Ideally, one mole of CH<sub>4</sub> consumed should remove one mole of oxygen from the solid in order to get  $2H_2$  and 1 CO. Thus, 100% selectivity should be reached if the lines of CH<sub>4</sub> consumed and O removed overlap each other. Ideally, they would follow the black line corresponding to the cumulated amount of CH<sub>4</sub> fed into the reactor and therefore the maximum CH<sub>4</sub> possibly consumed. If the consumption of CH<sub>4</sub> is higher than the one of O removed, some carbon deposition is expected. On the contrary, if the oxygen removed is higher than the one of CH<sub>4</sub> consumed, some total oxidation of CH<sub>4</sub> occurs producing H<sub>2</sub>O and CO<sub>2</sub> (besides or not some carbon deposition).

Fig. 12 shows such results obtained with three different catalyst amounts (50, 100 and 200 mg) at 700 °C. Except at the initial stage of the reductant step, methane is never totally consumed. With 50 mg of catalyst, the 0 removed curve seems to reach a maximum at approx. 55  $\mu$ mols. The amount of methane consumed also reaches a maximum but near to 75  $\mu$ mols. This is indicative of a limitation of the available oxygen for the reaction. With 100 mg, the overall O removed increases up to approx. 95  $\mu$ mols. Methane consumption on the other hand reaches 120  $\mu$ mols, thus indicating that carbon deposition occurs. With 200 mg, both curves are nearer to each other and increase evenly during the full step. Contrary to the experiments with lower amounts, the curves do not show tendencies to flatten due to the slowing down of O release or methane consumption.

It appears clearly that the reaction is limited by the amount of oxygen available. Interestingly, the process seems to slow down when the corresponding amount of oxygen needed to reduce  $\text{CeO}_2$  to  $\text{Ce}_6\text{O}_{11}$  is reached (42, 84, 168  $\mu$ mols for 50, 100 and 120 mg, respectively).

At  $750 \,^{\circ}\text{C}$  (Fig. 13), a similar trend can be observed. In this case the maximum values reach, naturally, higher values than at  $700 \,^{\circ}\text{C}$ . A deeper reduction of the solid is thus possible, well beyond the nominal  $\text{Ce}_6\text{O}_{11}$  composition. It can nevertheless be noted that

for both 50 and 100 mg experiments the curves of methane consumption and O removal tend to separate when this composition is reached.

These results suggest that the reduction of  $CeO_2$  to  $Ce_6O_{11}$  would be responsible for the rapid and selective conversion of  $CH_4$  to syngas, whereas more reduced ceria would be less reactive in terms of oxygen supply and thus lead to coke formation. Such consideration would be in accordance with the thermodynamic calculations mentioned in Section 3.1.

Nevertheless, XRD analysis of the catalysts after stopping the reaction after the reductant or oxidant steps, both at 600 and 800 °C did not reveal the presence of a  $Ce_6O_{11}$  phase in any case (Fig. 9). Thus, the actual phase transition to  $Ce_6O_{11}$  does not occur even when the corresponding nominal  $CeO_{1.83}$  composition is reached or even passed.

Fig. 14 shows the methane consumption and O removal during the reductant step using samples with different Ni loadings (2, 8.8 and 38.5 wt.%). For 2 wt.%Ni, the curve of O removal is above the one of CH<sub>4</sub> consumption, meaning that more oxygen is available in

the solid than that necessary to selectively oxidize the CH<sub>4</sub> which it is capable to activate. Going to the other extreme, high loading of 38.5 wt% of Ni considerably increases the amount of methane consumed as expected. On the other hand, the amount of oxygen consumed remains very similar to that of the 2 wt% loading. Most interestingly, the intermediate loading of 8.8 wt% shows an effect of Ni on both the surface activation of CH<sub>4</sub> and that of oxygen supply by the solid. This could also be explained by the presence of Ni species in strong interaction with Ce species with the eventual formation – in the intermediate composition range – of a Ce-Ni-O solid solution in which Ni<sup>2+</sup> species promote the oxygen mobility, and thus availability for DRM, of ceria. Indeed, such solid solutions are known for  $Ce_{1-x}Ni_xO_y$  with x < 0.5 [e.g. 37,57-59]. High loadings (38.5 wt% Ni, x = 1.86) would be prejudicial as independent phases can be created, forming majorly separated oxides such as NiO and CeO<sub>2</sub> during calcination. This NiO may lead to total oxidation during the first chemical looping cycles but then would remain in reduced state and contribute essentially to methane activation. On the contrary, a too low% of Ni (2 wt%, x = 0.06) can create a Ce-Ni-O

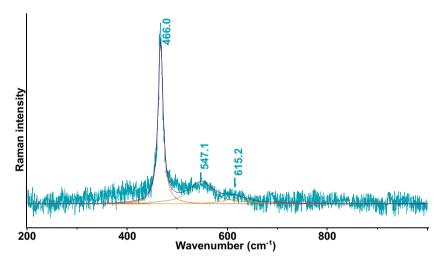


Fig. 11. Raman spectrum decomposition for Ni(9.5)/CeO<sub>2</sub> (800 °C, after CH<sub>4</sub>).

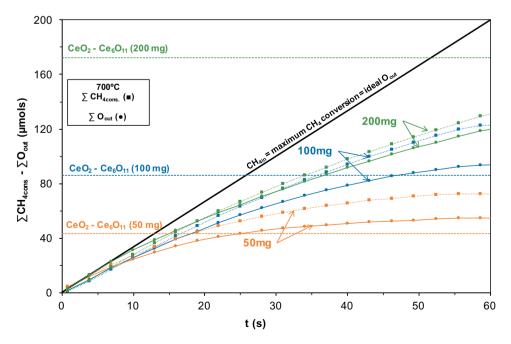


Fig. 12. O removed (●) and CH<sub>4</sub> consumed (■) at 700 °C on Ni(8.8)/CeO<sub>2</sub> with [CH<sub>4</sub>]<sub>0</sub> = 5% and after 11 cycles working in periodic.

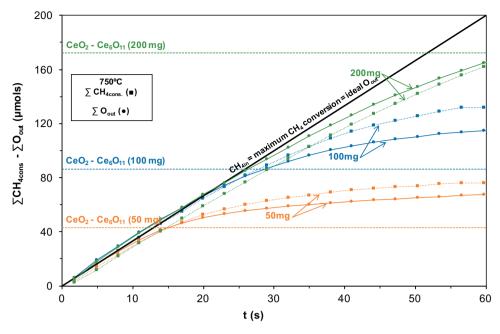


Fig. 13. O removed (●) and CH<sub>4</sub> consumed (■) at 750 °C on Ni(8.8)/CeO<sub>2</sub> with [CH<sub>4</sub>]<sub>0</sub> = 5% and after 11 cycles working in periodic.

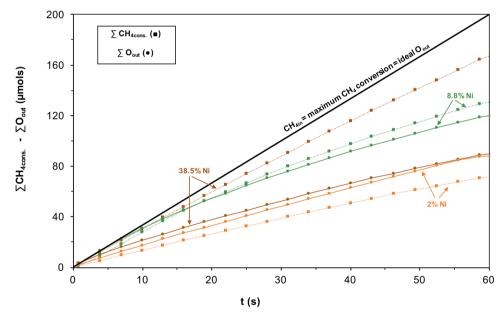


Fig. 14. O removed (●) and CH<sub>4</sub> consumed (■) at 700 °C on Ni(x)/CeO<sub>2</sub> (200 mg) with [CH<sub>4</sub>]<sub>0</sub> = 5% and after 11 cycles working in periodic.

solid solution but not in the optimum range (not enough oxygen vacancies). However, the 9-10 wt% loading of Ni (x = 0.28-0.35) can create a solid solution near to the optimum on the studied series.

Characterization of the various solids did not reveal the presence of such solid solution. Nevertheless, both XPS and Raman scattering suggest the presence of Ni<sup>2+</sup> species in strong interaction with Ce species. It can therefore be supposed that such species play an important role on the oxygen mobility in the system either in the bulk of the support or, more probably, at its surface. In this respect Ni loading, and thus distribution of these species at the surface is a crucial parameter for the optimization of such process.

#### 4. Conclusions

The principle of periodic dry reforming of methane has been successfully demonstrated using a solid based on particles of nickel

(Ni) supported on cerium dioxide (CeO<sub>2</sub>). The support reacts as an oxygen vector by reduction and oxidation under CH<sub>4</sub> and CO<sub>2</sub> respectively. The metal is necessary to provide the good activation of methane, leading to high conversions. Bulk metallic nickel is not oxidized back to the original oxide state in presence of CO<sub>2</sub> under the working conditions, which is also corroborated with the thermodynamic calculations, whereas surface species may be oxidized by CO<sub>2</sub>. In such conditions, the selectivity coming from the oxidation of methane is practically ideal towards syngas, on the contrary to when the catalyst is re-oxidized by oxygen. In addition, the main differences from other studies made in the literature are the high selectivity towards H<sub>2</sub>/CO without a high production of H<sub>2</sub>O with Ni/CeO<sub>2</sub> instead of Fe-CeO<sub>2</sub> formulations and the ease of the synthesis technique. Impregnation method, the most practical one for the industry, has shown promising performances for this process. Results also showed that carbon formed during exposure

to methane can be fully removed during the re-oxidation by CO<sub>2</sub>. This allows to avoid carbon accumulation and deactivation.

From the dynamic study and the characterization of catalysts, it has been shown that Ni species play two essential roles. First, they contribute to the activation of the reactants as expected. Second, Ni also has an important role on the rate of oxygen supply from the solid, especially in the intermediate loading showing an important synergetic effect that does not appear at very low and very high loadings. This could be linked to the strong interaction of part of the surface nickel species with cerium species, and a promotion of oxygen mobility from the support to active Ni particle and backwards.

Obviously, other important aspects regarding the reactivity of such systems need to be studied such as, i.e., effect of reactant and in particular methane partial pressures or the re-oxidation process of both the oxide and of carbon. Nevertheless, this work opens the path to further optimization, in particular by:

- The control of the amount and the nature of the metallic phase, see mixed metals phase to optimize the activation of CH<sub>4</sub> and CO<sub>2</sub>.
- The control of the oxygen available to react in the oxygen carrier modifying the nature of the support (oxides, mixed oxides...)

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#### Appendix A. Supplementary data

Supplementary data associated with this article can be found, in the online version, at http://dx.doi.org/10.1016/j.apcatb.2017. 04.048.

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